

Disorder and Magnetism: An Examination of Some Spinel and Perovskite Systems

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Spinel: The motivation [Brent Melot]

Spinel CuMn_2O_4 [Daniel Shoemaker]

Perovskite $(\text{Ba}/\text{Sr})(\text{Ti},\text{Nb})\text{O}_3$ [Katharine Page]

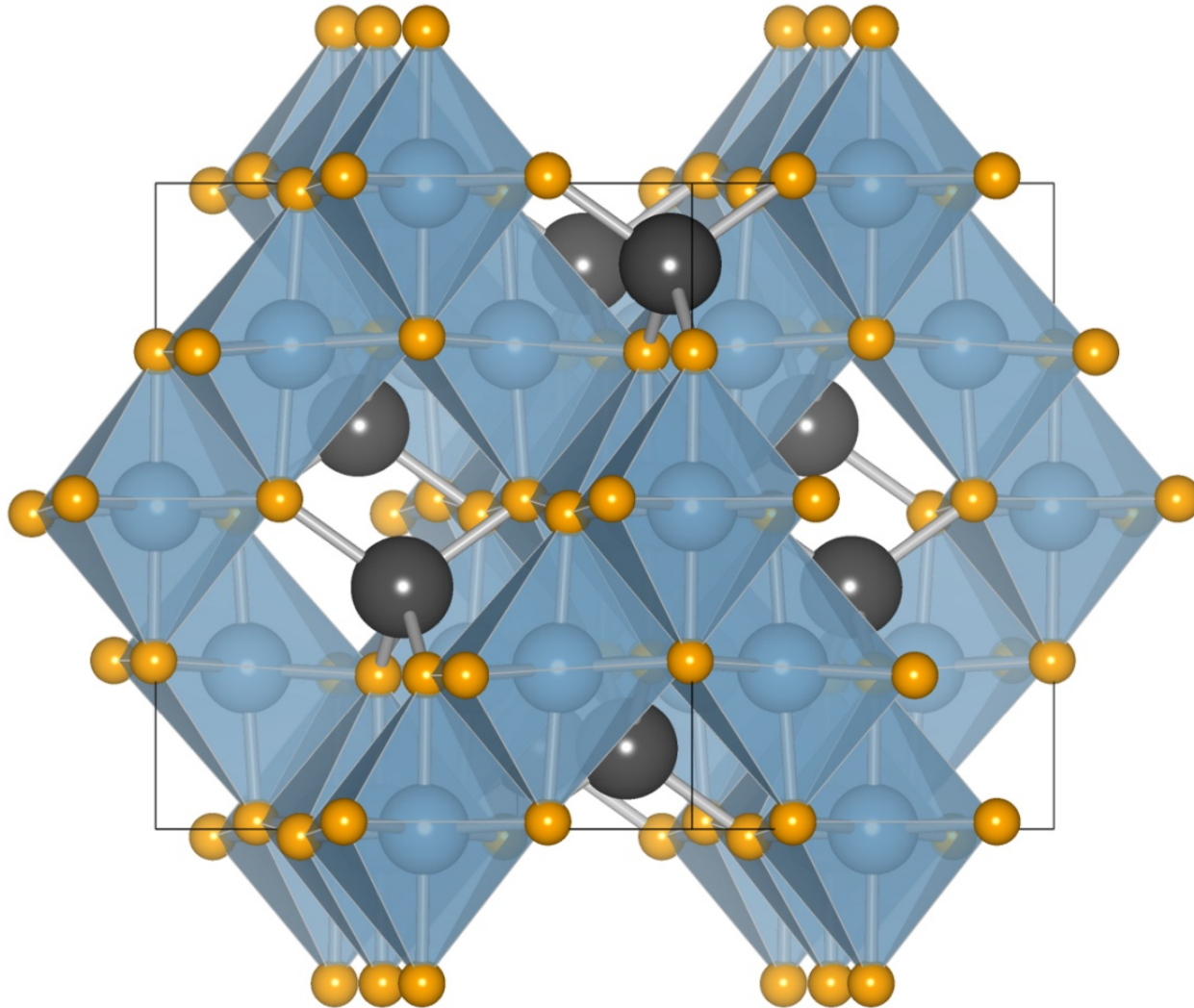
Funding: NSF Career, Institute for Multiscale Materials Studies,
UCSB ICMR, NSF GSF to KP.



Spinels: The Motivation

Thanks: A. P. Ramirez, G. Lawes

The spinel structure



Edge-shared AO_6 octahedra, with tetrahedral B ions connecting the octahedra.

A-B is the principle magnetic interaction, and it is usually uncompensated antiferromagnetic.

B-B and A-B are also important

The spinel structure

The AB_2O_4 palette: Almost all ions are closed magnetic shells.

Exceptions: ■ Strong *B*-site tendency: (s)

A	Magn.	JT	B	Magn.	JT	Aniso.
Mn ²⁺	Y	N	V ³⁺ (s) ■	Y	Y	N/Y
Fe ³⁺	Y	N	Cr ³⁺ (s)	Y	N	N
Co ²⁺	Y	N	Mn ³⁺ (s) ■	Y	Y	N/Y
Ni ²⁺ ■	Y	Y	Fe ²⁺	Y	N	Y
Cu ¹⁺	N	N	Fe ³⁺ ■	Y	N	Y
Cu ²⁺ ■	Y	Y	Co ²⁺	Y	N	Y
Zn ²⁺	N	N	Co ³⁺	N	N	N
Ga ³⁺	N	N	Rh ³⁺ (s)	N	N	N
			Ni ²⁺	Y	N	N
			Cu ²⁺ ■	Y	N	N
			Al ³⁺	N	N	N

Insulating oxide spinel magnets: New magnetoelectrics

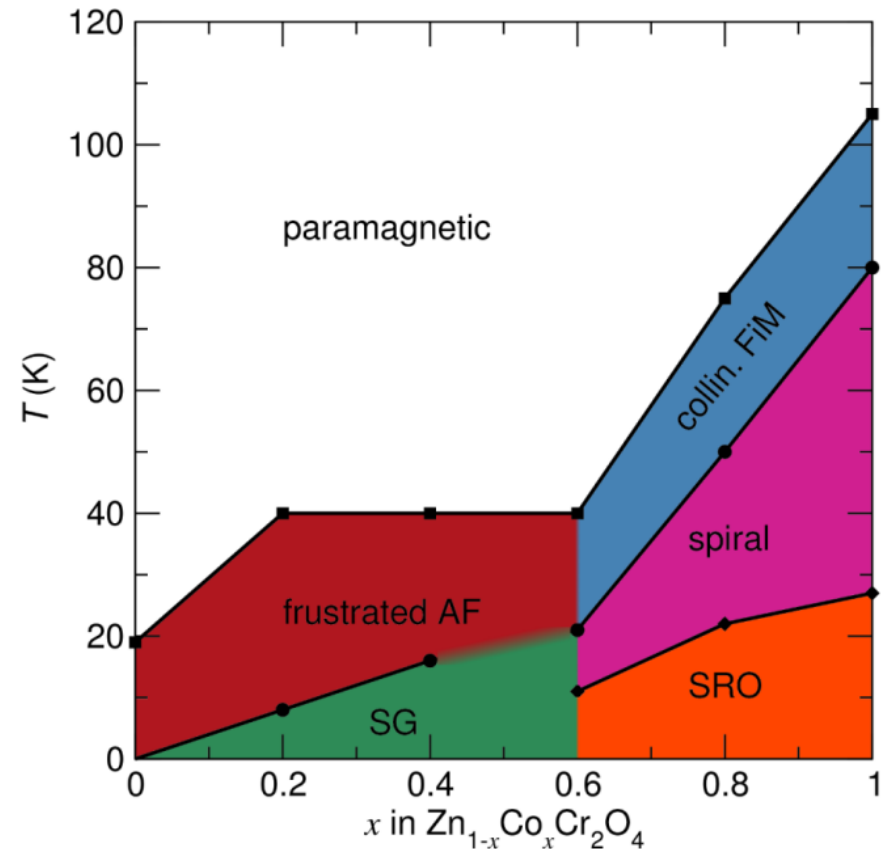
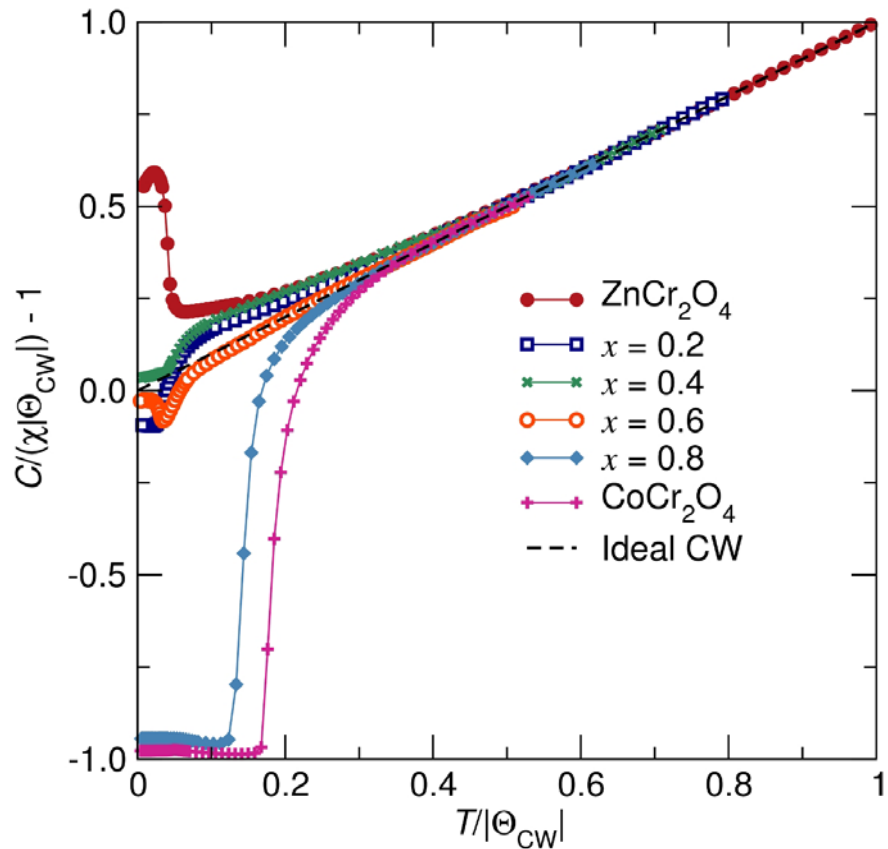
CoCr_2O_4 : Lawes, Melot, Page, Ederer, Proffen, Hayward, Seshadri, *Phys. Rev. B* **74** (2006) 024413(1-6).

Mn_3O_4 : Tackett, Lawes, Melot, Grossman, Toberer, Seshadri, *Phys. Rev. B* **76** (2007) 024409(1-6).

Inhomogeneous magnetism (exchange biasing)

ZnMn_2O_4 : Shoemaker, Rodriguez, Seshadri, Abumohor, Proffen, *Phys. Rev. B* **80** (2009) 144422(1-9).

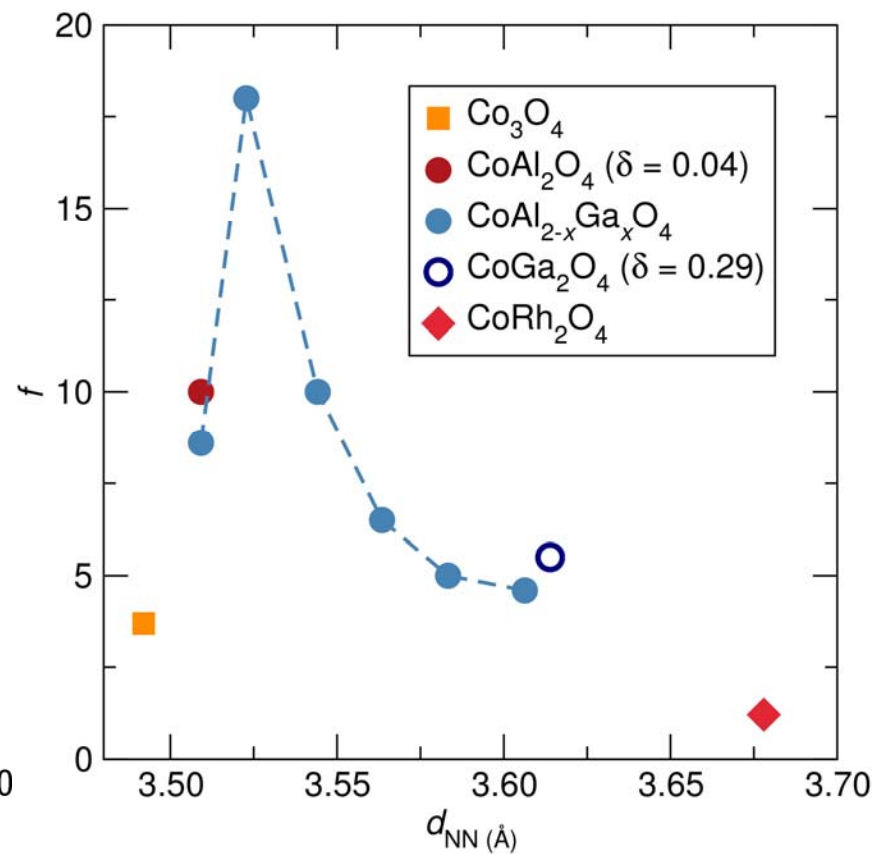
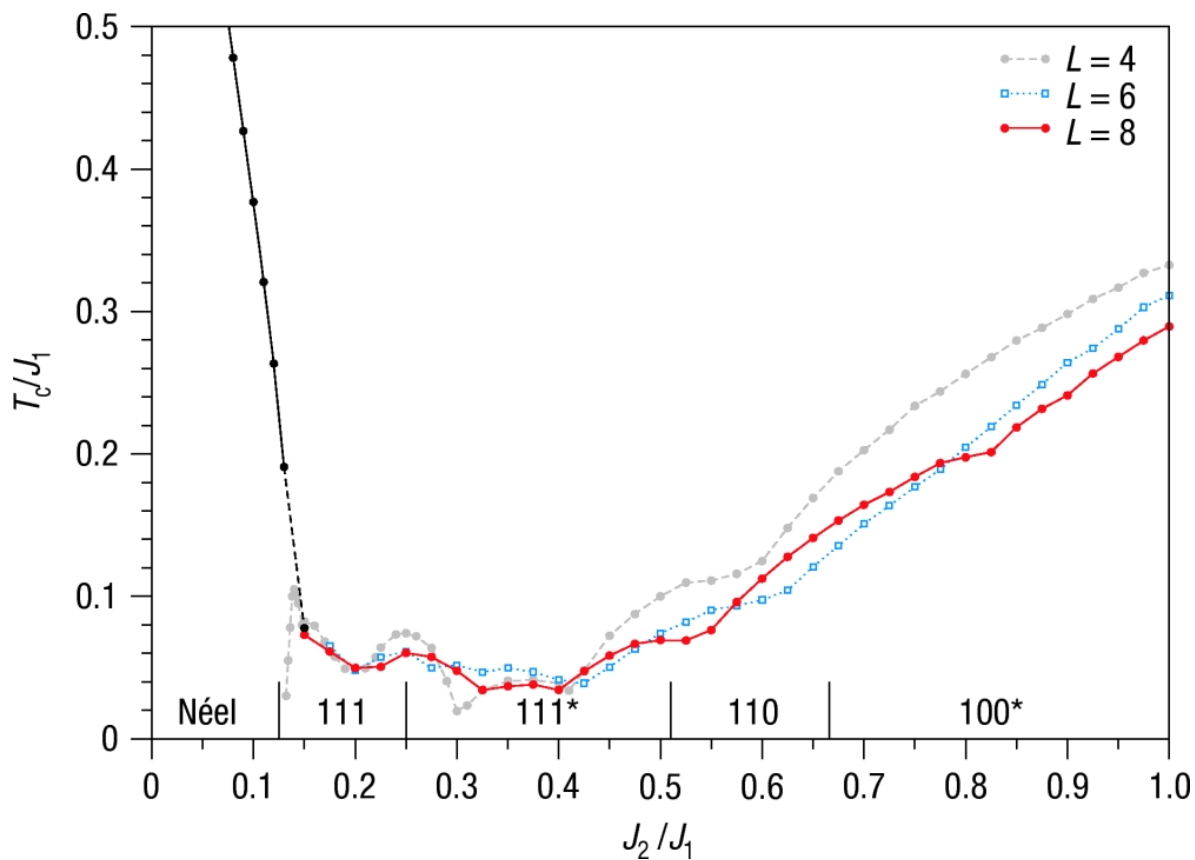




Rich magnetic phase diagram obtained by increasing A-B magnetic interactions whilst preserving B-B. "*Morphotropic*" *magnetolectrics*?

Melot, Drewes, Seshadri, Stoudenmire, Ramirez,
J. Phys. Condensed Matter 21 (2009) 216007(1-7).

CoAl_{1-x}Ga_xO₄: A site-only magnetism



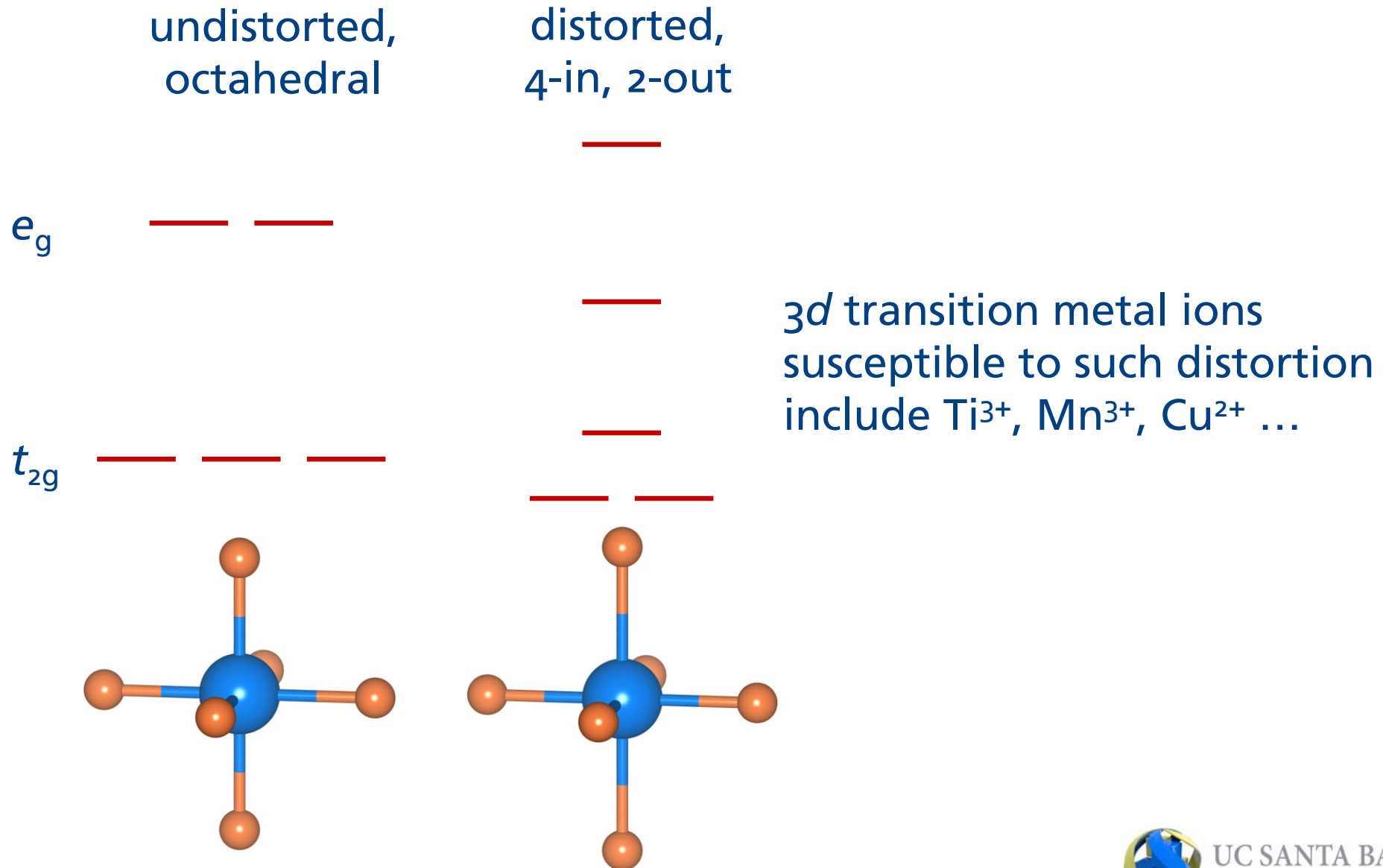
Bergman, Alicea, Gull, Trebst, Balents,
Nature Phys. **3** (2007) 487-491.

Melot, Page, Seshadri,
Stoudenmire, Balents,
Bergman, Proffen, *Phys. Rev. B.* **80** (2009) 104420(1-8).

Disorder in Jahn-Teller-active CuMn_2O_4

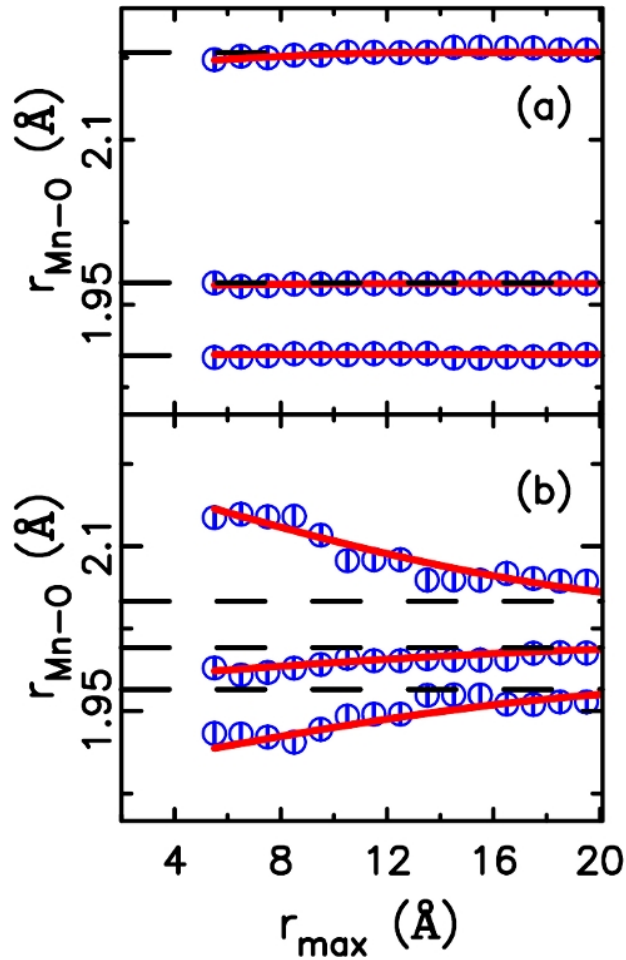
Shoemaker, Li, Seshadri, *J. Am. Chem. Soc.* **131** (2009) 11450–11457.

Disorder in Jahn-Teller-active CuMn_2O_4

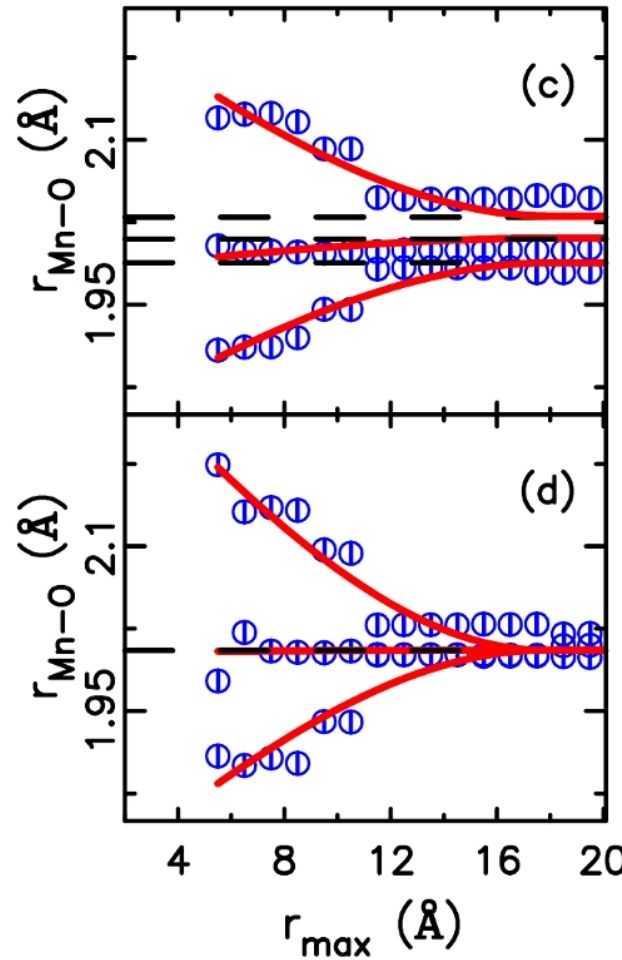


Jahn-Teller effects in LaMnO_3

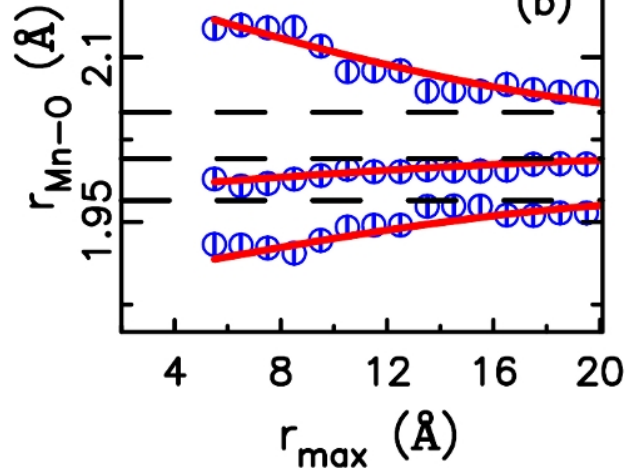
300 K



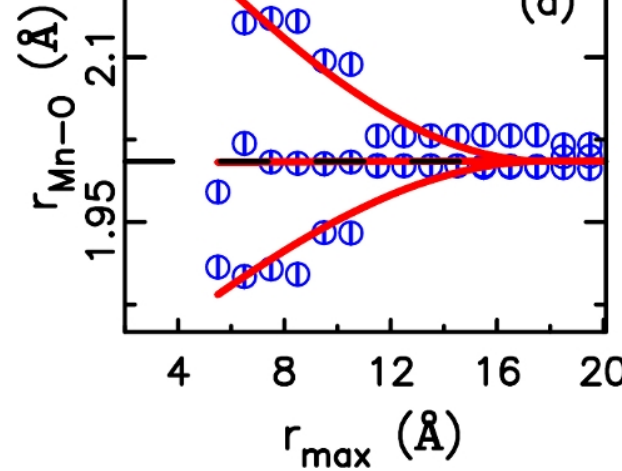
800 K



740 K

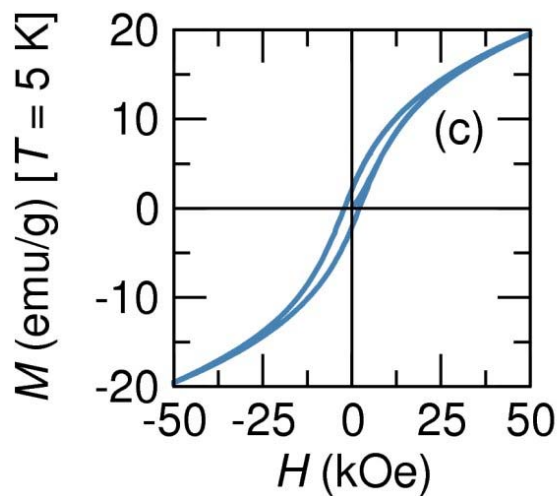
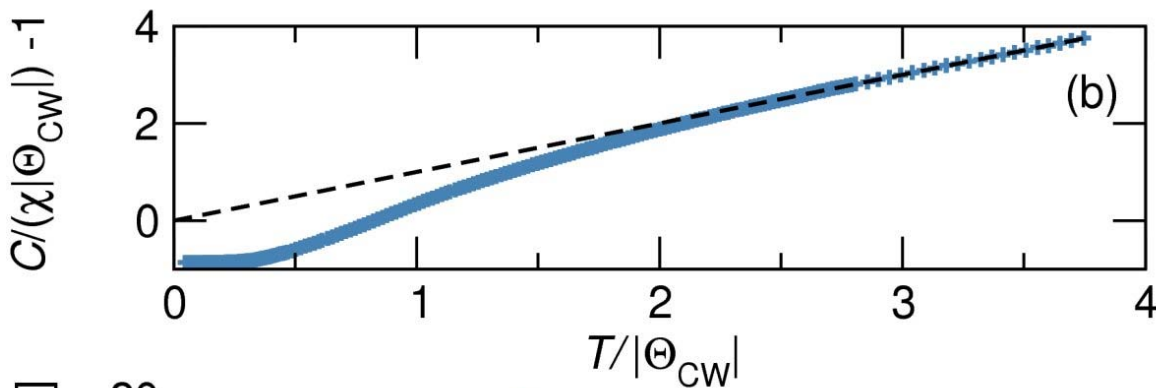
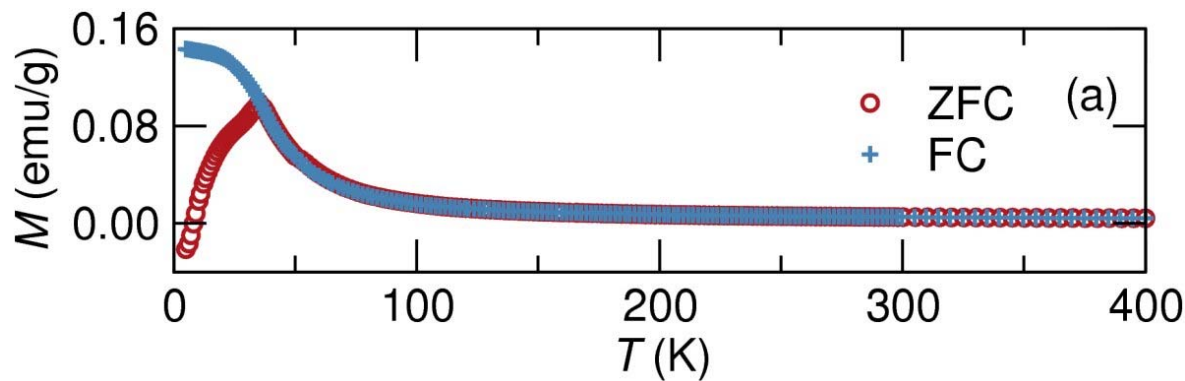


1100 K



The example of LaMnO_3 shows the need for *local* probes for understanding cooperative Jahn-Teller: Qiu, Proffen, Mitchell, Billinge, *Phys. Rev. Lett.* 94 (2005) 177203(1-4).

Disorder in Jahn-Teller-active CuMn_2O_4

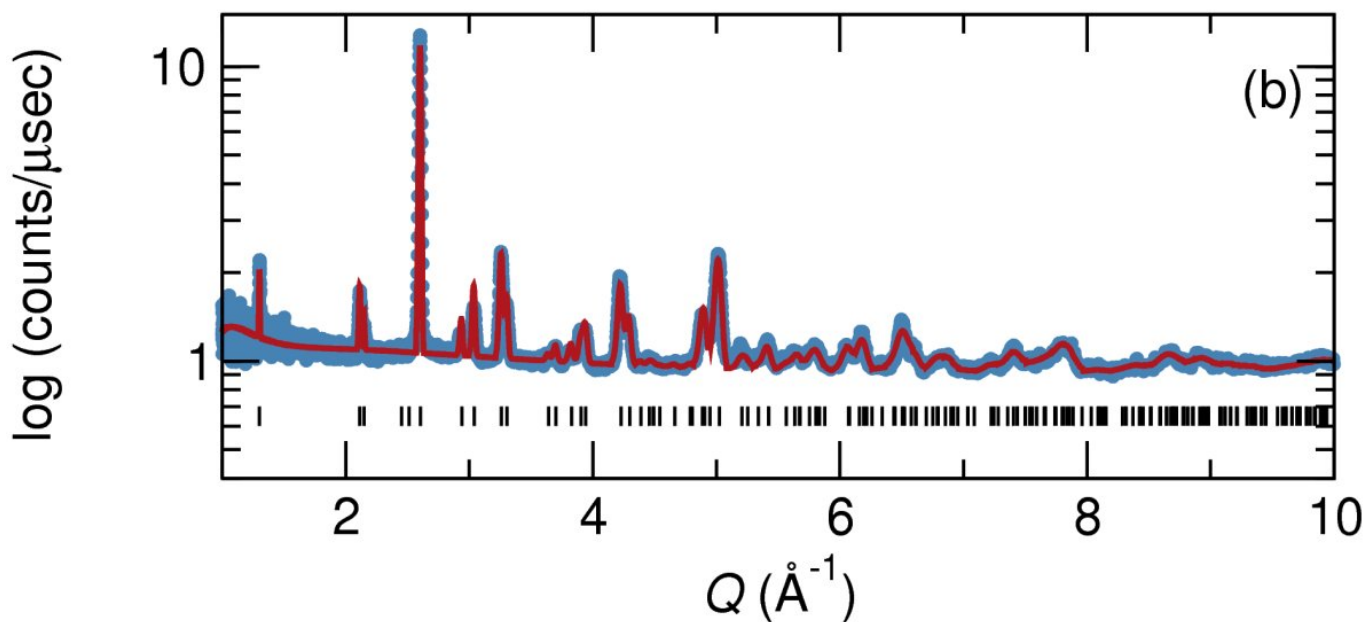
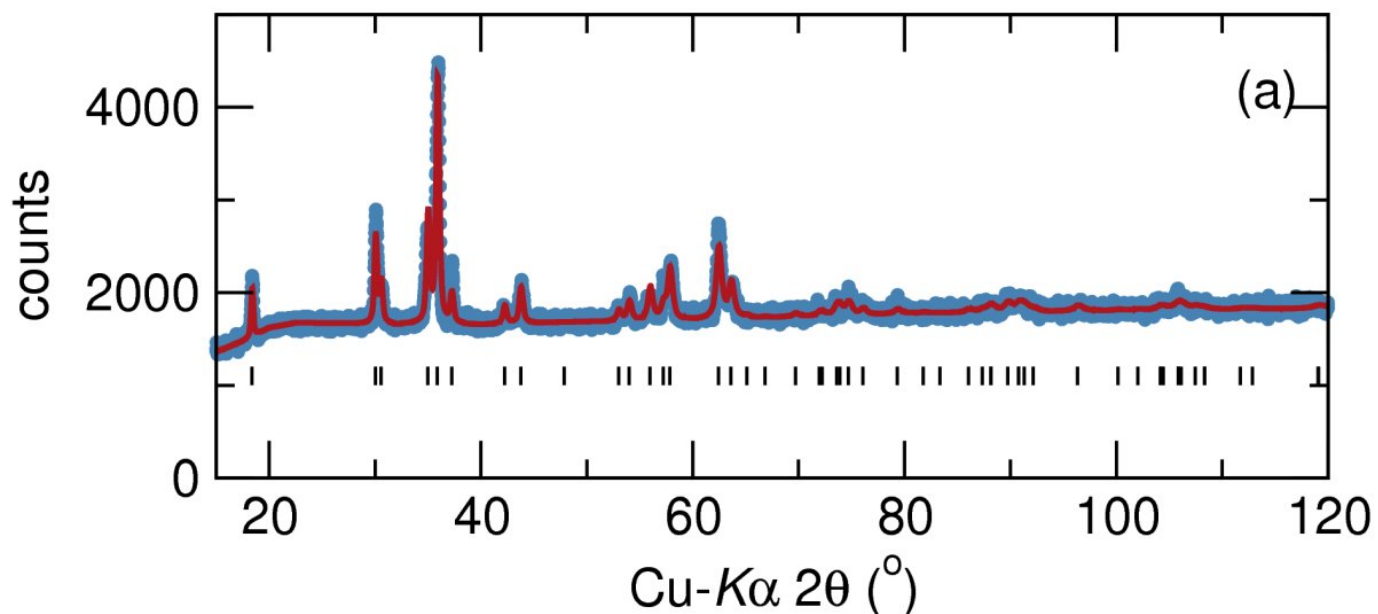


Magnetization acquired under a field of 100 Oe, showing the emergence of a hard ferrimagnet below 40 K.

$$\Theta_{\text{CW}} = -107 \text{ K.}$$



Disorder in Jahn-Teller-active CuMn_2O_4



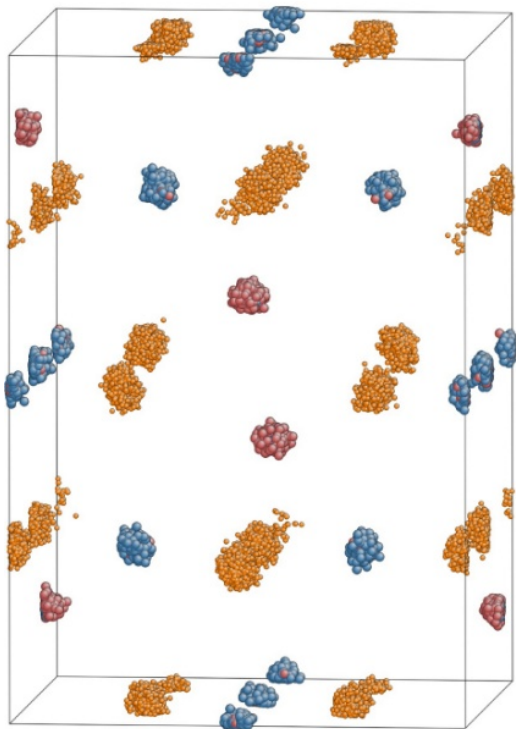
A metastable tetragonal spinel with two Jahn-Teller ions. Site mixing as well (close to 30%).

Note the rapid drop in scattering (lab X-ray and NPDF, Los Alamos) intensity, even for neutrons.

$I_{41/amd}$ structure does a good job in describing the data.

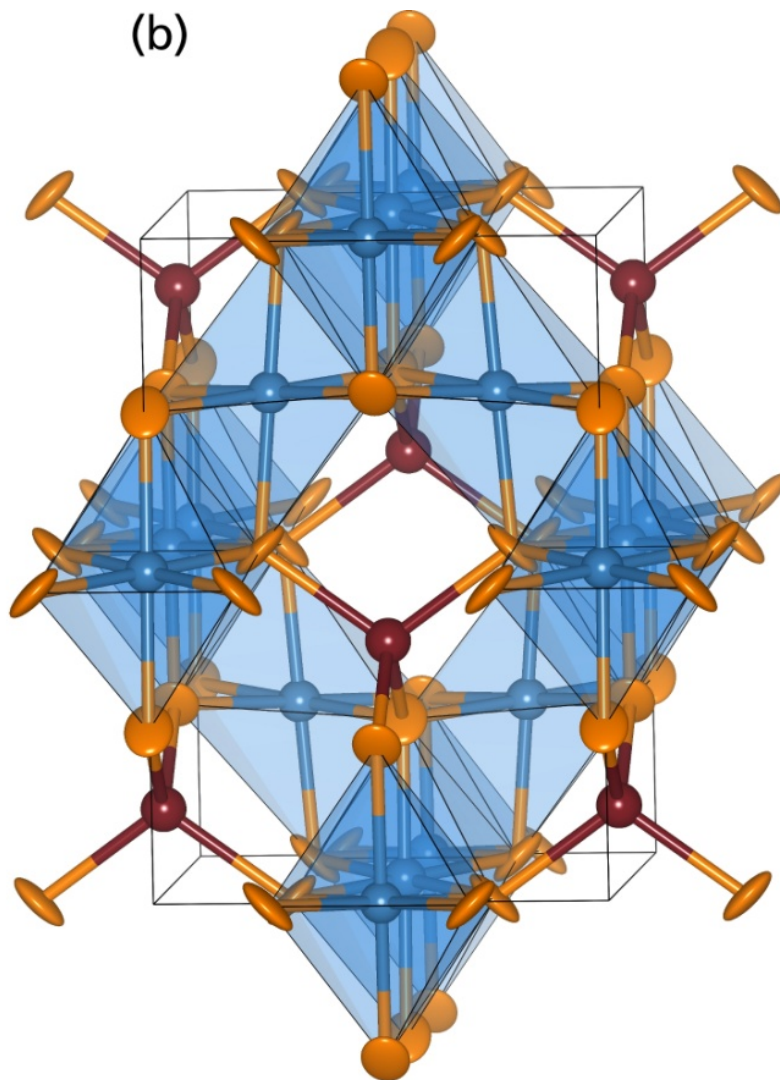
Disorder in Jahn-Teller-active CuMn_2O_4

(a)



$10 \times 10 \times 7$

(b)

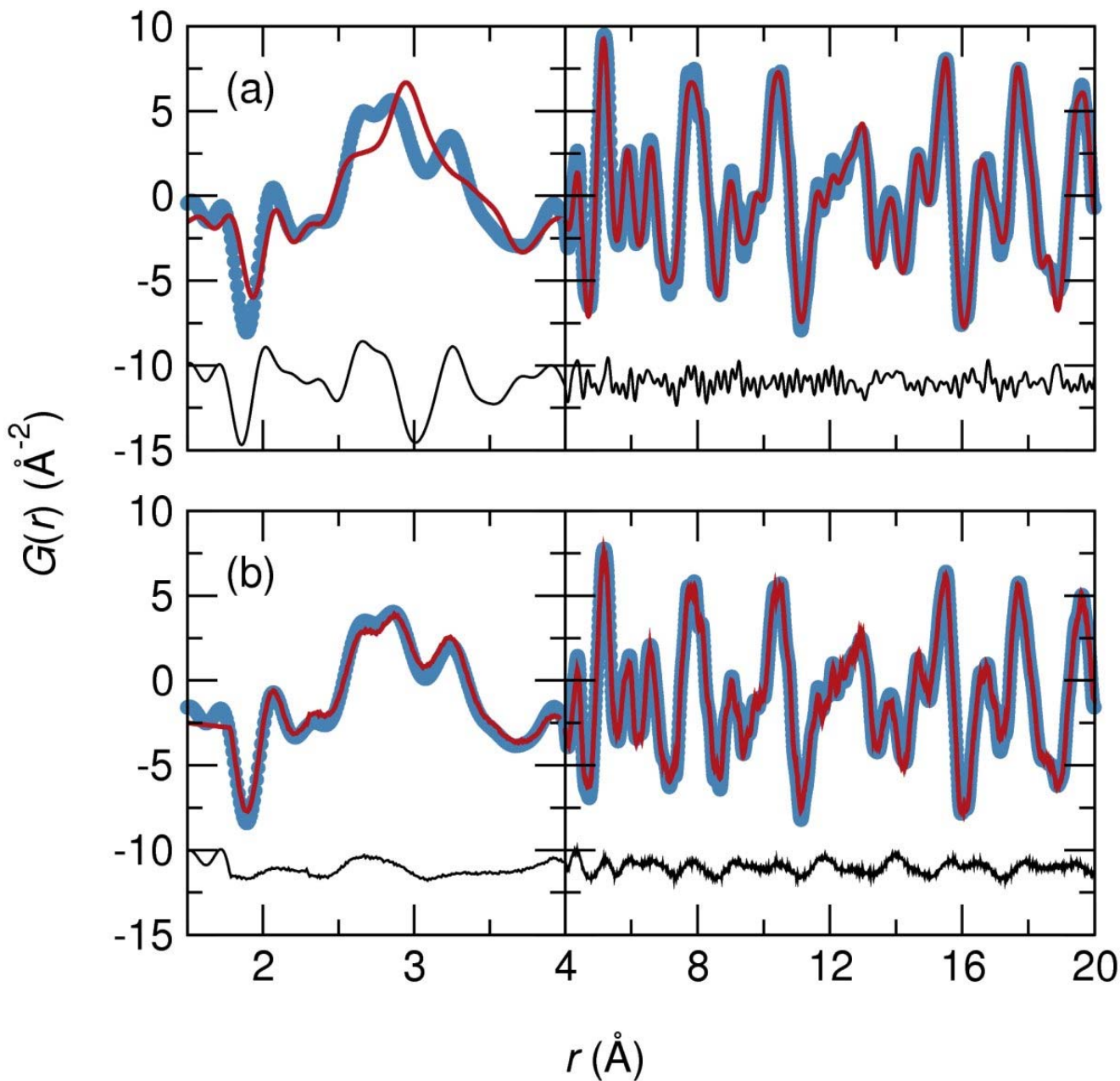


RMC modeling of the total scattering $G(r)$, using a between 20,000 and 40,000 atom models; ~ 6 million steps. Hops between A and B sites allowed.

Note smearing of atom positions and contrast the anisotropic thermal parameters from Rietveld.

RMCProfile: Tucker, Keen, Dove, Goodwin, Hui, *J. Phys.: Condens. Matter* **19** (2007) 335218.

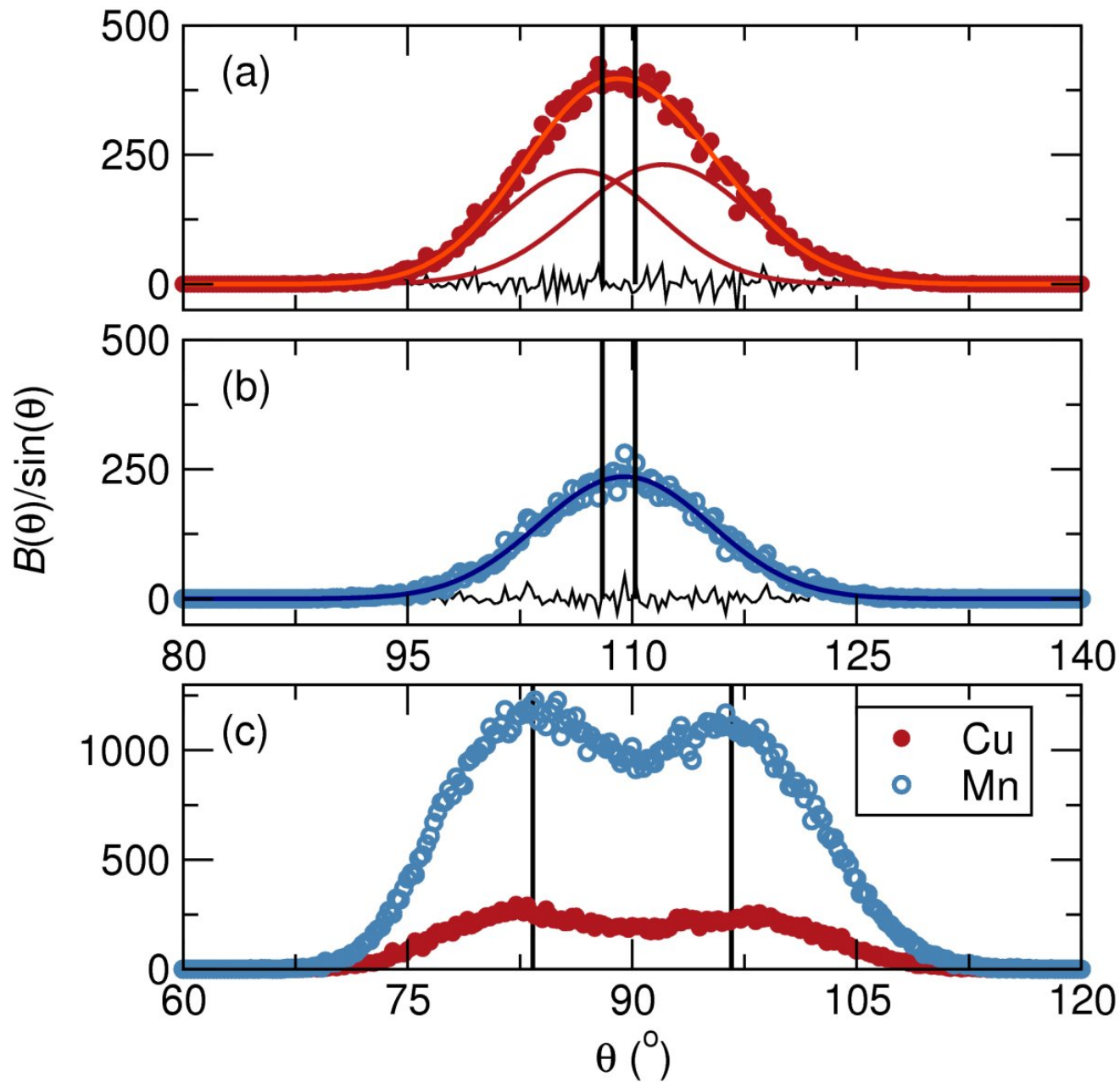
Disorder in Jahn-Teller-active CuMn_2O_4



The Rietveld (average) structure fits the $G(r)$ well beyond the second-near neighbors.

RMC does a much better job all over.

Disorder in Jahn-Teller-active CuMn_2O_4

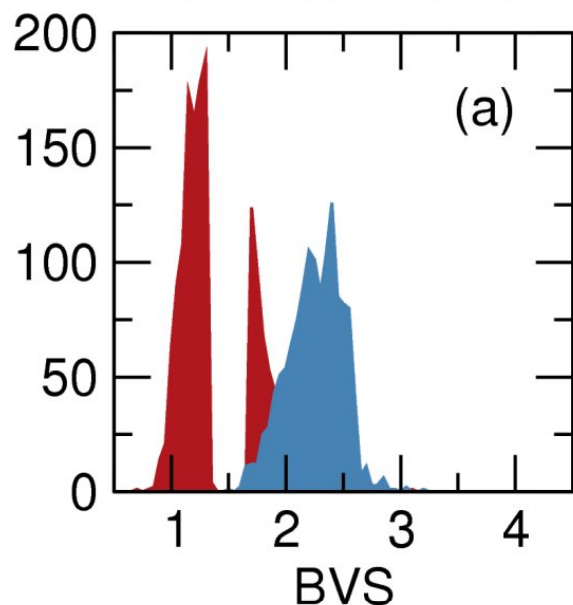


What does a JT active ion do on a tetragonal site?

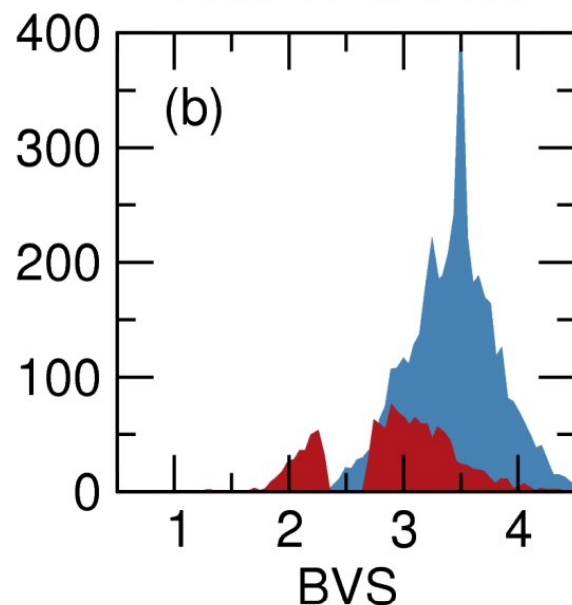
Cu^{2+}O_4 tetrahedra appear flattened, with deviations from $109^\circ 28'$

Disorder in Jahn-Teller-active CuMn_2O_4

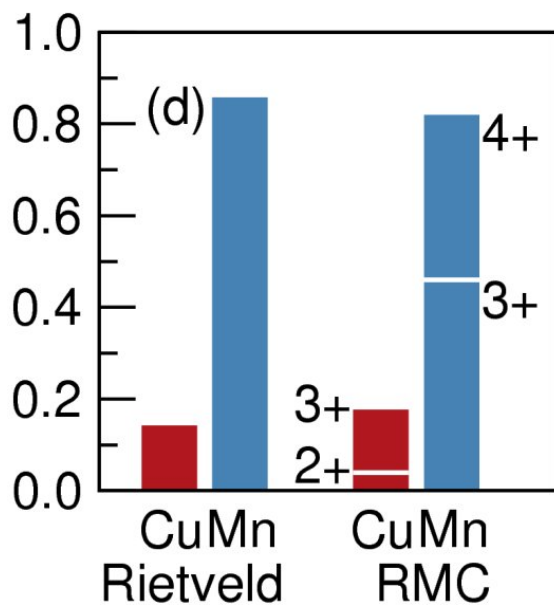
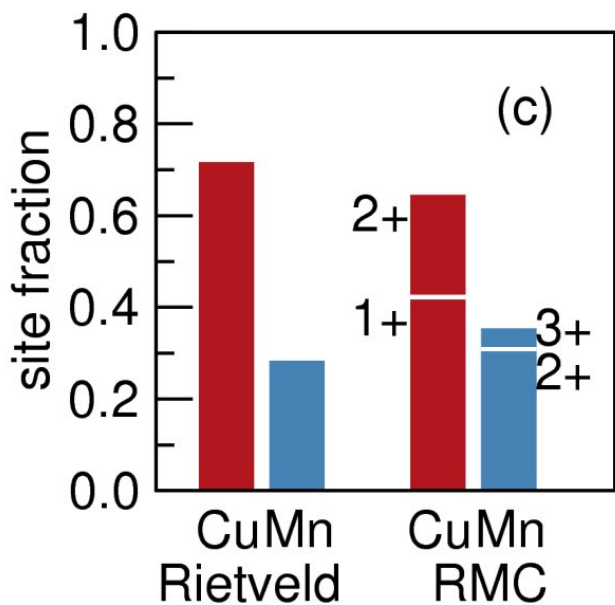
tetrahedral sites



octahedral sites



The explicit atom-by-atom description allows **bond valence sums** to be employed to determine oxidation state, separately on the two sites.

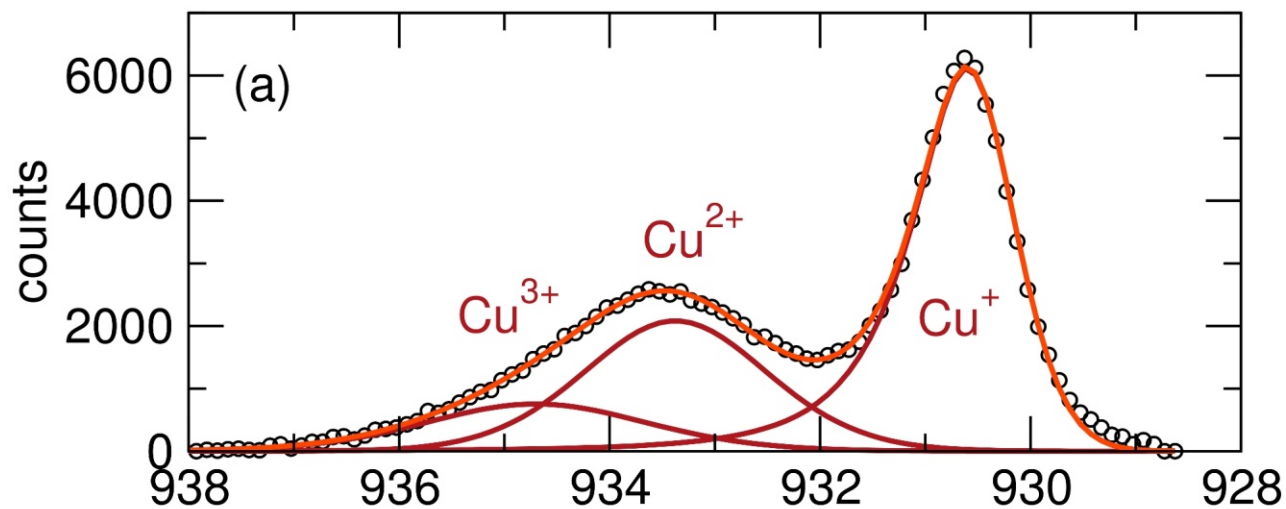


The big surprise: Cu^{3+}

The system disproportionates charge to avoid Jahn-Teller:

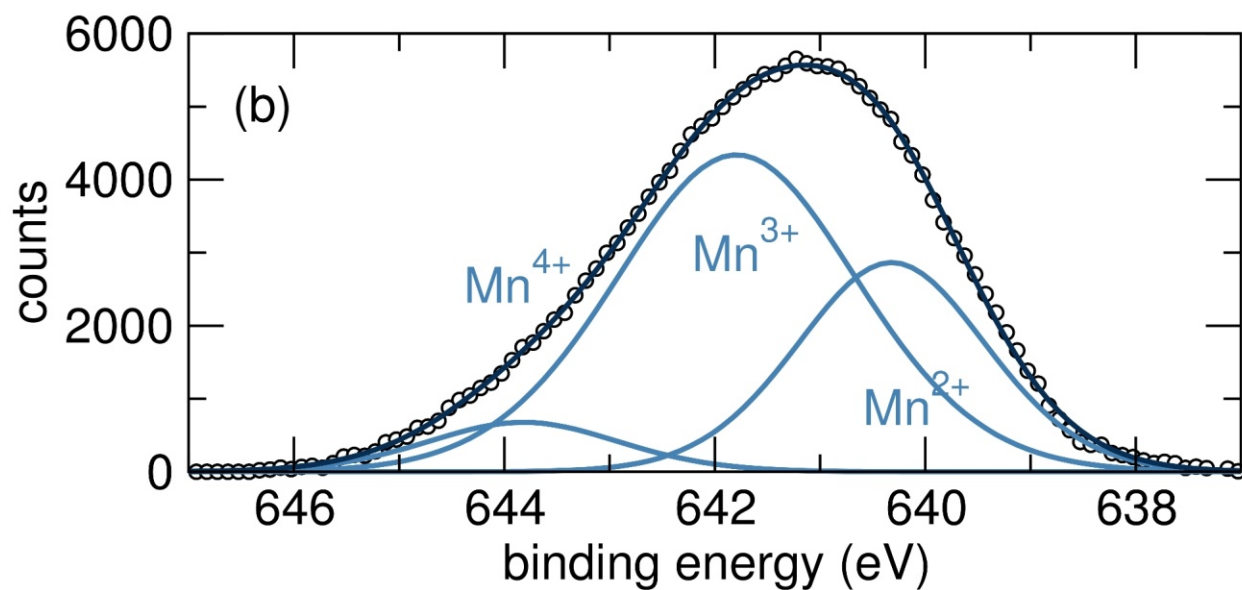


Disorder in Jahn-Teller-active CuMn_2O_4



XPS shows up Cu^{3+} as well.

Assignment not previously made.



Disorder in Jahn-Teller-active CuMn_2O_4

	Cu			Mn		
charge	tet.	oct.	XPS	tet.	oct.	XPS
1+	0.42 (42%)		47%			
2+	0.20 (20%)	0.10 (10%)	35%	0.34 (17%)		35%
3+		0.28 (28%)	18%	0.05 (2%)	0.80 (44%)	47%
4+					0.72 (36%)	18%

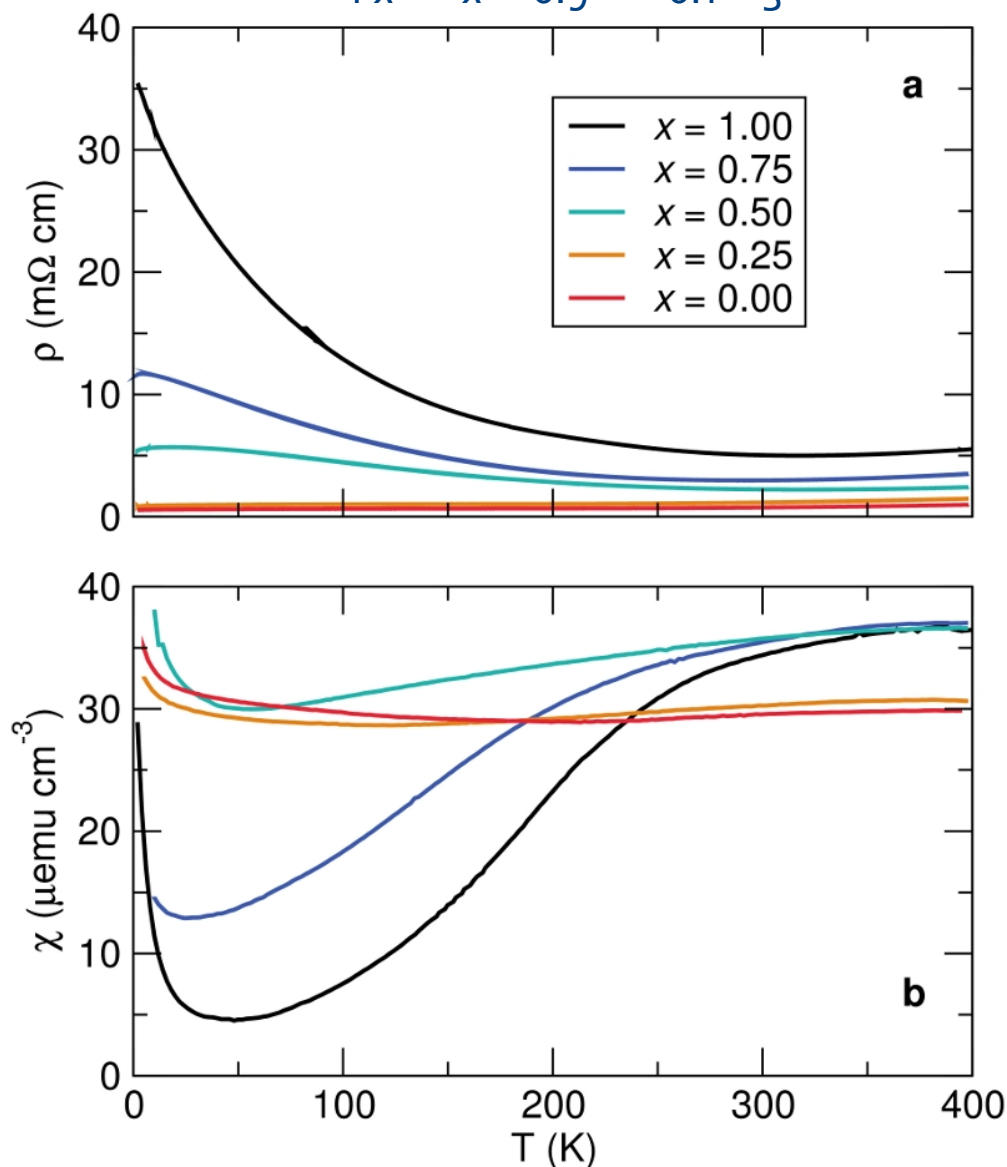
2 sites, 2 cations, and 4 oxidation states.

Charge disproportionation seems to be driven by a tendency to avoid Jahn-Teller states (Cu^{2+} and Mn^{3+} on the tetrahedral site)

Perovskite (Ba/Sr)(Ti,Nb)O₃

Page, Kolodiazhnyi, Proffen, Cheetham, Seshadri, *Phys. Rev. Lett.* **101**
(2008) 205502.

Perovskite (Ba/Sr)(Ti,Nb)O₃



SrTiO_3 and BaTiO_3 are band insulators.

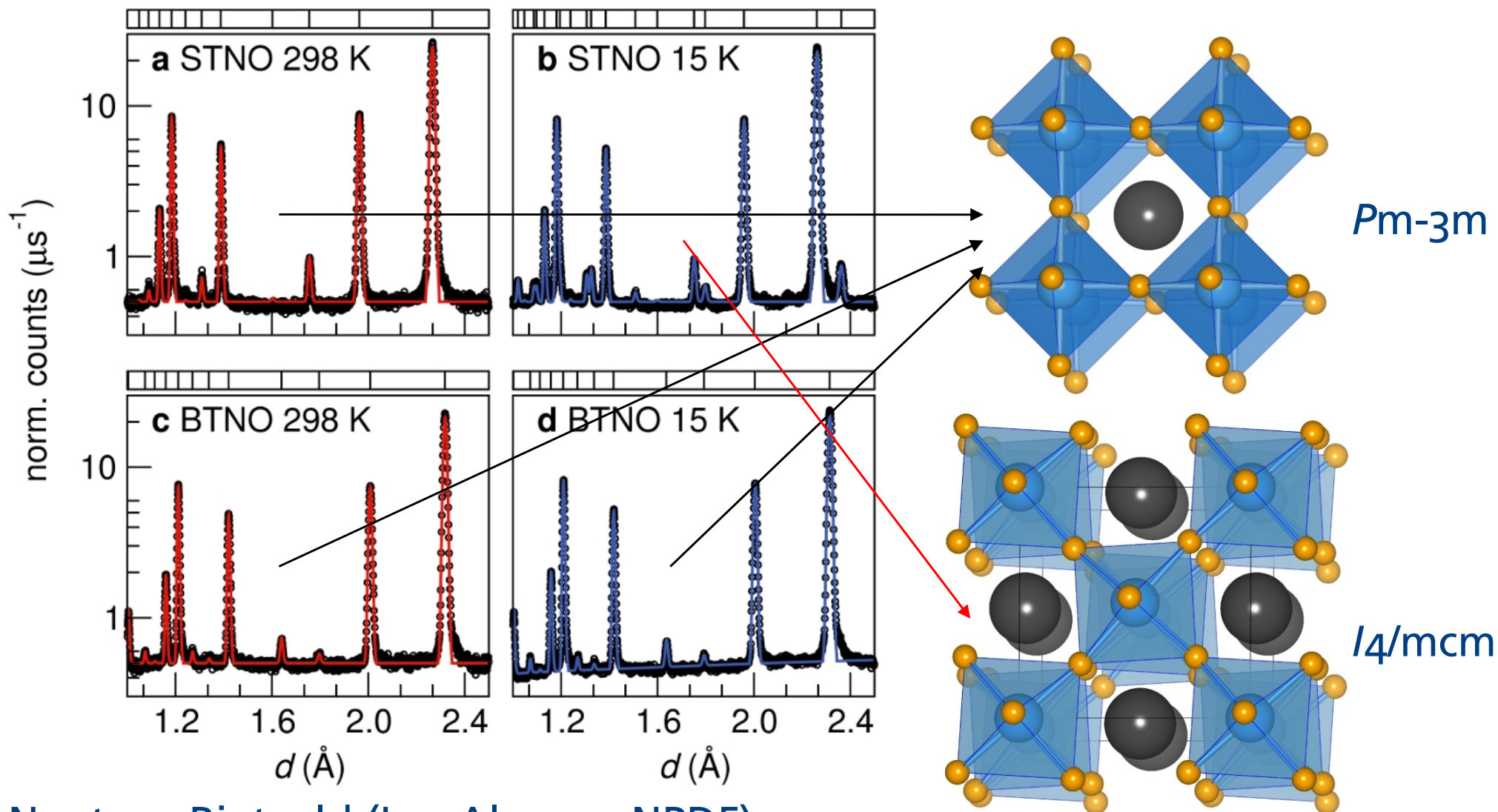
Nb^{4+} -substitution (even small amounts) make SrTiO_3 metallic and even superconducting.

Not so BaTiO_3

Samples from T. Kolodiazhnyi, NIMS, Tsukuba

Perovskite (Ba/Sr)(Ti,Nb)O₃

What does structure tell us? Compare SrTi_{0.875}Nb_{0.125}O₃ (STNO) and BaTi_{0.875}Nb_{0.125}O₃ (BTNO): 1/8th substitution to aid modeling

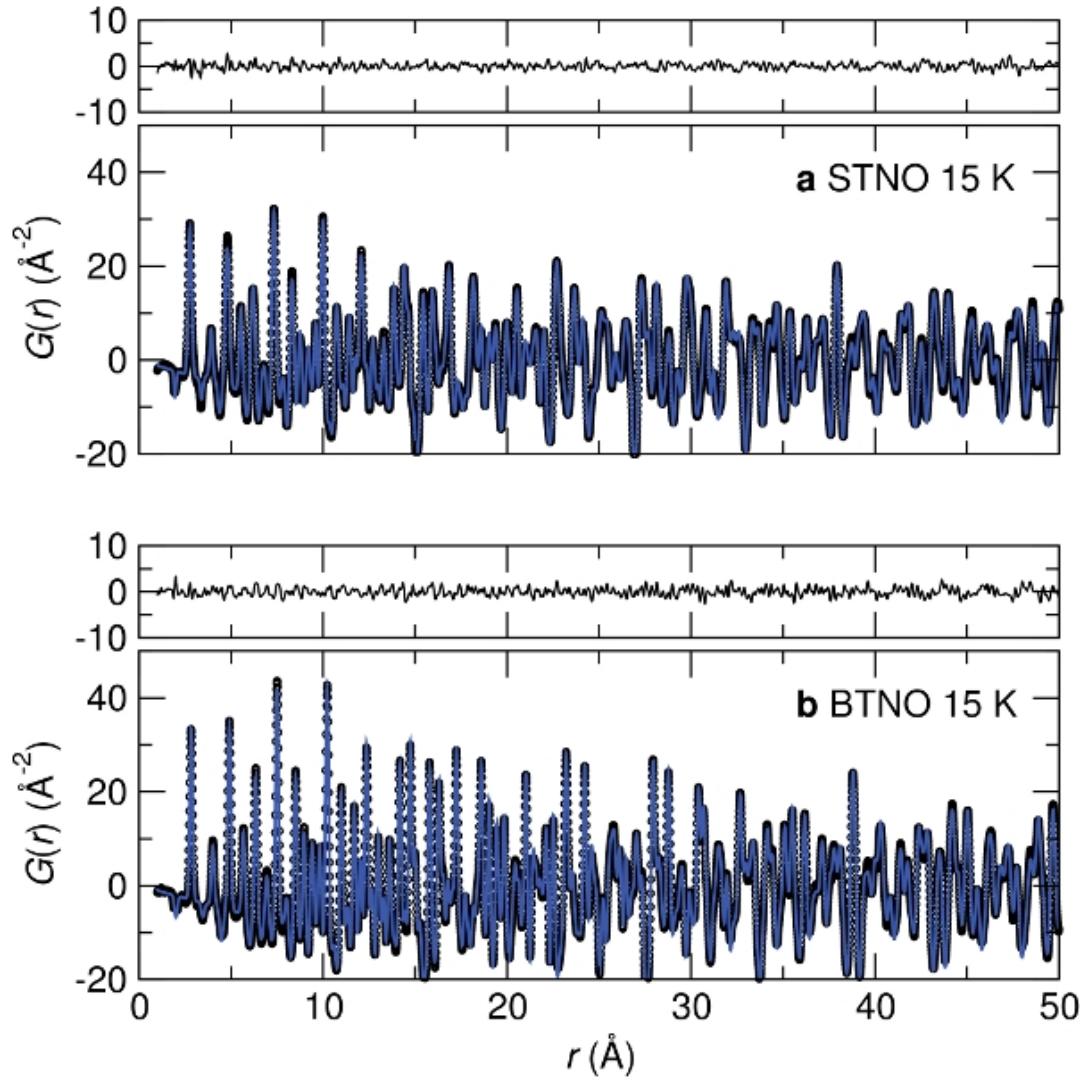


Neutron Rietveld (Los Alamos, NPDF)

KITP 2010

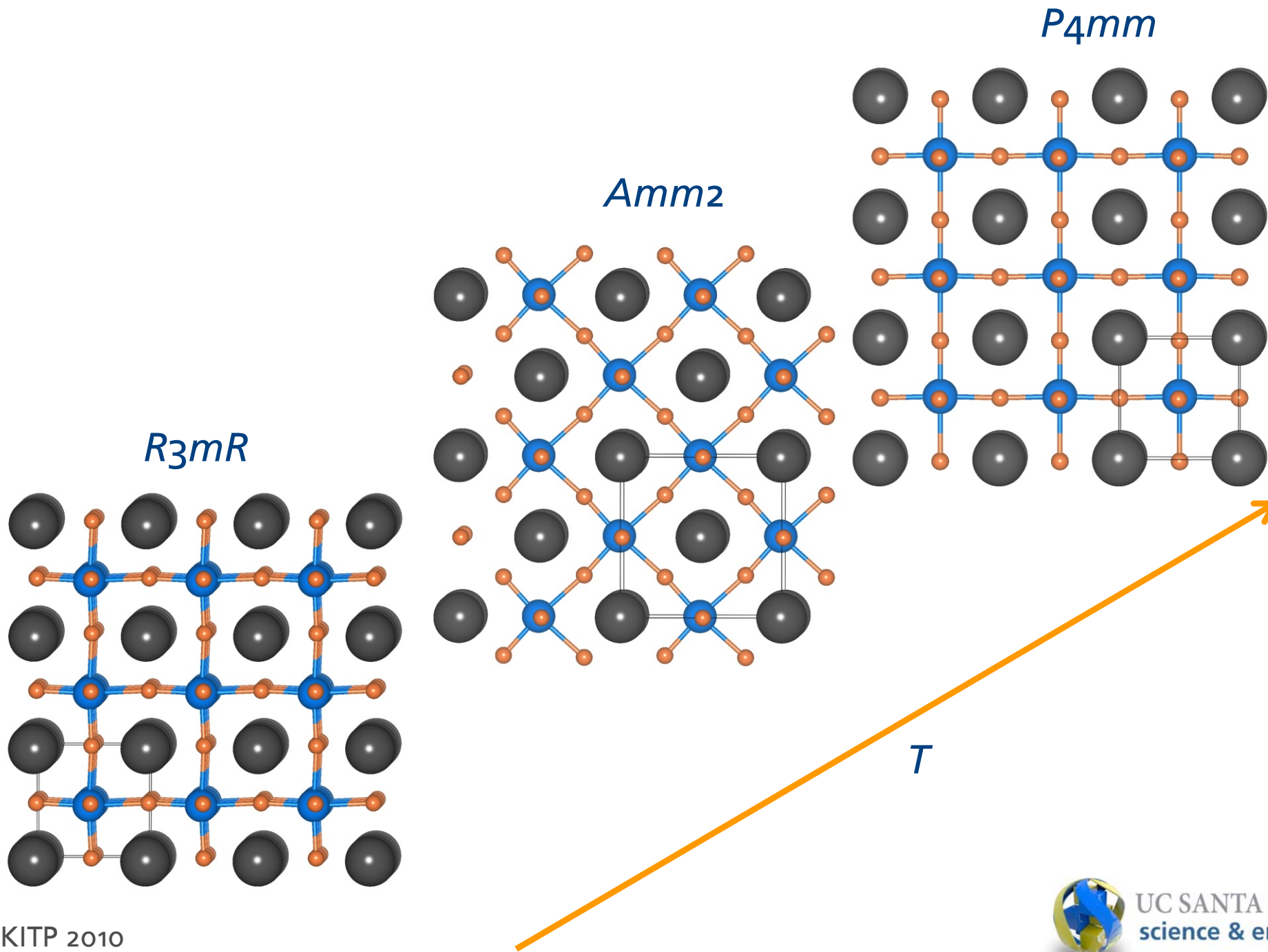
Perovskite (Ba/Sr)(Ti,Nb)O₃

Pair distribution functions fit the average structure out to rather long vectors, very well !



The implication is that the average structure is a good approximation of the real structure.

Perovskite (Ba/Sr)(Ti,Nb)O₃



Perovskite (Ba/Sr)(Ti,Nb)O₃

However ...

	SrTi _{0.875} Nb _{0.125} O ₃		15 K, <i>I4/mcm</i>	
	300 K, <i>Pm</i> $\bar{3}$ <i>m</i>			
	Rietveld	PDF	Rietveld	PDF
<i>a</i> (Å)	3.9237(1)	3.9255(1)	5.5391(2)	5.5418(4)
<i>c</i> (Å)			7.8312(4)	7.835(1)
<i>U</i> _{iso} (Sr)	0.0098(1)	0.0076(2)	0.0053(1)	0.0034(1)
<i>U</i> _{iso} (Ti/Nb)	0.0056(3)	0.0050(2)	0.0042(3)	0.0028(1)
<i>U</i> _{iso} (O)	0.0098(1)	0.0084(1)	0.0054(1)	0.0042(1)
<i>x</i> (O2)			0.2349(1)	0.2350(1)
Occupancy (Nb)	0.127(2)	1/8	0.123(2)	1/8
<i>R</i> _w (%)	3.6	7.7	3.8	7.7

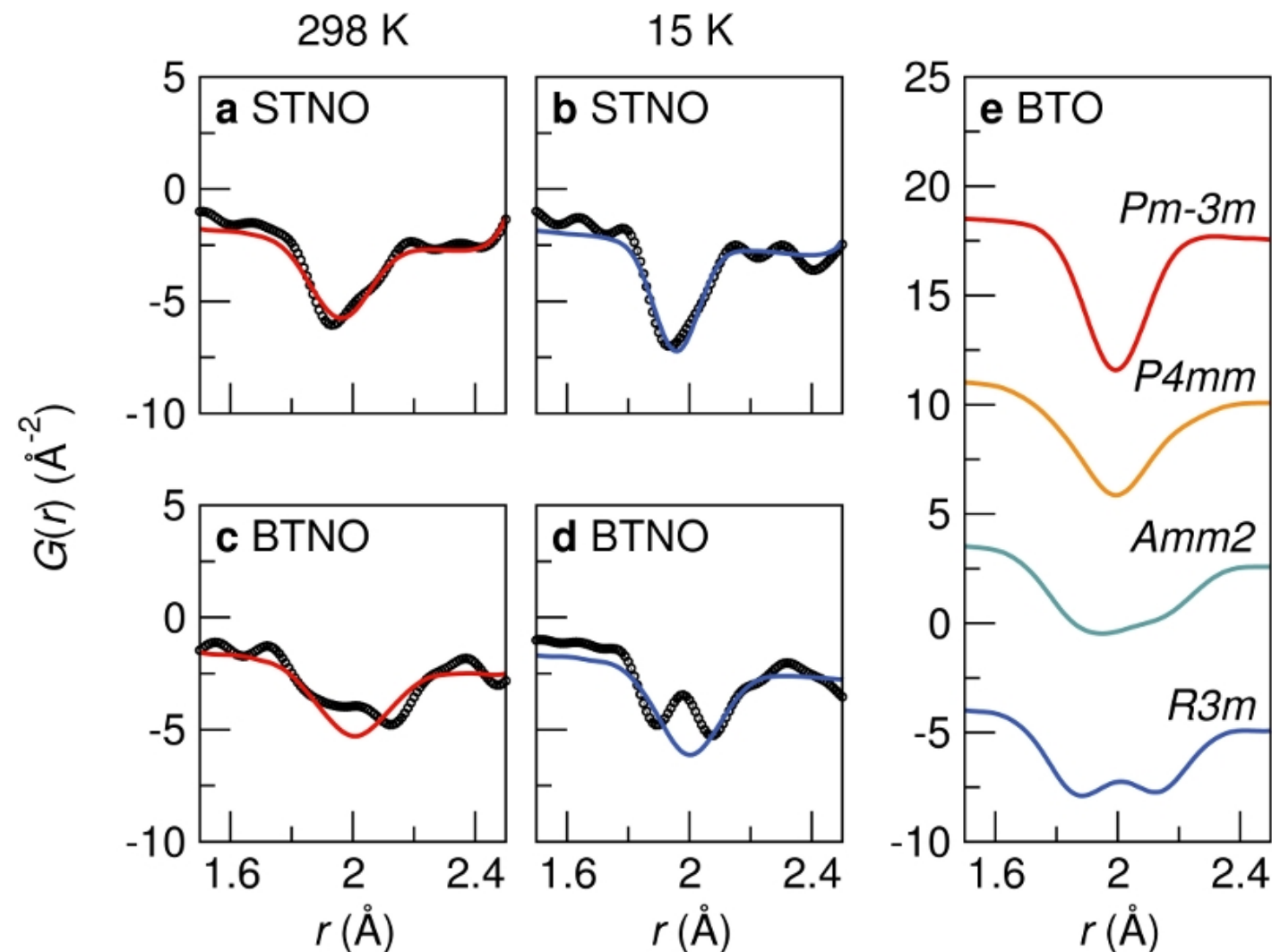
Suggest a tendency to tilt

	BaTi _{0.875} Nb _{0.125} O ₃		15 K, <i>Pm</i> $\bar{3}$ <i>m</i>	
	300 K, <i>Pm</i> $\bar{3}$ <i>m</i>			
	Rietveld	PDF	Rietveld	PDF
<i>a</i> (Å)	4.0147(1)	4.0165(1)	4.0084(1)	4.0100(1)
<i>U</i> _{iso} (Ba)	0.0061(1)	0.0052(3)	0.0028(1)	0.0021(1)
<i>U</i> _{iso} (Ti/Nb)	0.0093(3)	0.0080(2)	0.0070(2)	0.0059(1)
<i>U</i> _{iso} (O)	0.0077(1)	0.0061(1)	0.0053(1)	0.0040(1)
Occupancy (Nb)	0.131(1)	1/8	0.132(1)	1/8
<i>R</i> _w (%)	3.0	7.4	3.4	8.5

Suggests a tendency to off-center

Perovskite (Ba/Sr)(Ti,Nb)O₃

What about the very short range ? BaTi_{0.875}Nb_{0.125}O₃ displays distortions in the first (Ti/Nb)-O neighbor:



Simulations of the first peak of BaTiO₃ for the different crystal structures.

Perovskite (Ba/Sr)(Ti,Nb)O₃

Findings:



Nb-substitution (on the perovskite *B* site) does not frustrate tilting. The ground state is tetragonal (with enhanced tilting compared with SrTiO₃).

Tilting distortions *do not* influence the ground state, which is metallic (Cf. SrRuO₃, LaNiO₃ ...)



Nb-substitution (on the perovskite *B* site) frustrates long-range ordering of dipoles; additionally helped by dipole-dipole screening due to charge carriers. Average structure is cubic.

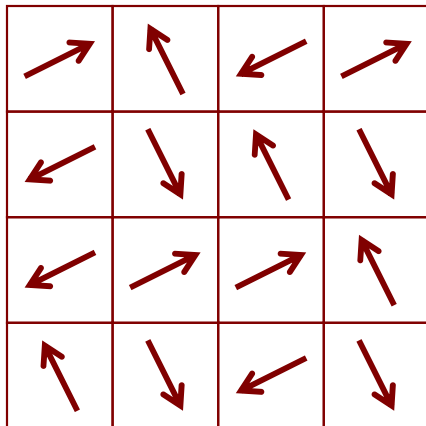
Dipoles exist locally however. Incoherent off-centering might aid the insulating ground state.



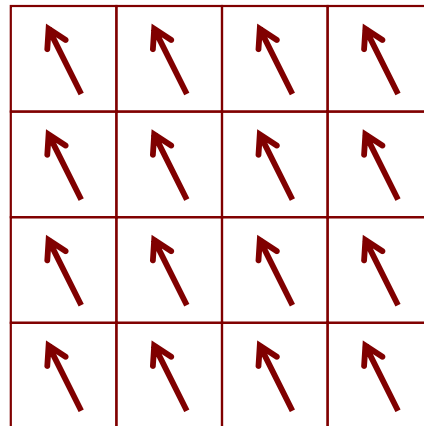
Summary

In many functional materials, modes of disordering exist which may or may not be coherent, and when they are not, special tools are required to see them.

incoherent

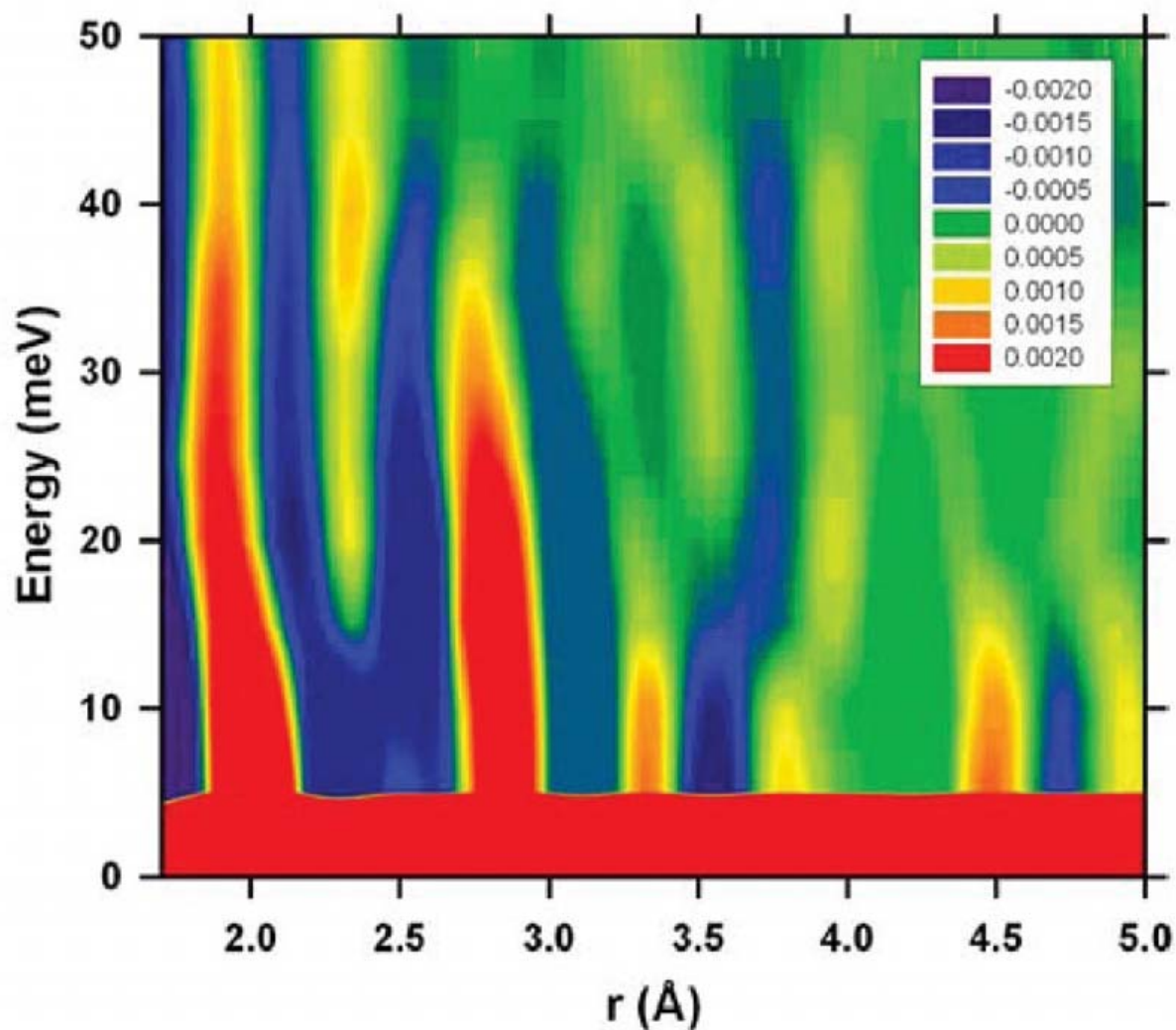


coherent



Local Lattice Dynamics and the Origin of the Relaxor Ferroelectric Behavior

W. Dmowski,¹ S. B. Vakhrushev,² I.-K. Jeong,³ M. P. Hehlen,⁴ F. Trouw,⁴ and T. Egami^{1,5,6}



Dynamic PDF:
 $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3$,
Pharos
diffractometer at
Los Alamos, 100 g
sample, 1 day per
temperature.

